

## SEMIEMPIRICAL RELATION BETWEEN NON-LINEAR SUSCEPTIBILITY (REFRACTIVE INDEX), LINEAR REFRACTIVE INDEX AND OPTICAL GAP AND ITS APPLICATION TO AMORPHOUS CHALCOGENIDES

H. Tichá, L. Tichý<sup>a</sup>

University of Pardubice, 532 10 Pardubice, Czech Republic

<sup>a</sup>Joint Laboratory of Solid State Chemistry of the Academy of Sciences of Czech Republic and the University of Pardubice, 532 10 Pardubice, Czech Republic

Using generalized Miller's rule and linear refractive index ( $n$ ), simple relation is suggested for estimation of the non-linear susceptibility ( $\chi^{(3)}$ ) and non-linear refractive index ( $n_2$ ). This relation provides reasonable agreement between the calculated and experimental  $\chi^{(3)}$ ,  $n_2$  values for more than 90 glassy oxides ( $\chi^{(3)}$ ) and also for 22 optical crystals ( $n_2$ ). Prediction of  $\chi^{(3)}$  and  $n_2$  values for chalcogenide glasses indicates that non-linear susceptibility can reach up to  $\chi^{(3)} \cong 2 \times 10^{-10}$  esu for some telluride glasses. Simple semiempirical correlation between  $n_2$  and the optical gap is suggested.

(Received April 15, 2002; accepted May 15, 2002 )

*Keywords:* Amorphous chalcogenides; Non-linear refractive index

### 1. Introduction

Amorphous (a) chalcogenides possess interesting optical properties: (i) high refractive index  $n(\text{a-GeS}_2) \cong 2.2$  [1],  $n(\text{a-As}_2\text{Te}_3) \cong 3.6$  [2], (ii) photoinduced changes manifested by blue or red (bleaching, darkening) shift of the optical gap ( $E_g$ ), e.g. [3], (iii) photoinduced crystallization, e.g. [4], (iv) photoinduced anisotropy, e.g. [5]. Amorphous chalcogenides are also transparent in the mid-infrared spectral region. For example bulk selenide glasses show good transparency up to 11  $\mu\text{m}$  if the content of oxygen impurities is of order ppm or less, see e.g. [6,7]. Various sulphide and selenide glasses are promising hosts of rare-earths for photonic applications as fibre lasers, optical amplifiers and upconvertors, see e.g. [8,9]. There are some experimental indications that amorphous or glassy chalcogenides have rather high non-linear optical susceptibility, see e.g. [10-14], and, hence, these materials are promising candidates as non-linear optical elements. The experimental determination of the non-linear optical properties of a material requires rather an elaborate technique, for example Z scan method [15], four-wave mixing [16] or optical third harmonic generation [10]. These methods are not common in most of laboratories, and, hence, of interest are empirical or semiempirical relations to predict non-linear refractive index ( $n_2$ ) or non-linear optical susceptibility ( $\chi^{(3)}$ ) from some linear optical constants of a material.

In this brief communication we shall use some simple semiempirical relations relating  $\chi^{(3)}$  to  $n$  in oxide glasses for prediction of  $\chi^{(3)}$  and  $n_2$  in amorphous chalcogenides.

### 2. The choice of semiempirical formula and its application

Several semiempirical formulas were suggested for calculation of  $\chi^{(3)}$  or  $n_2$  [17-25]. Among formulas presented here [17-25], most widely used in the field of oxide glasses seems to be the formula (38) introduced by Boling et al. [23] which relates  $n_2$  to linear refractive index ( $n$ ) and to the Abbe number. Important is the formula (7) introduced in [25] which relates  $n_2$  to the optical gap

( $n_2 \sim E_g^{-4}$ ) and which can predict  $n_2$  values for both the wide-gap dielectrics and semiconductors, too. Hajto et al.[26] used Boling's formula (38), see Ref. [23], for estimation of  $n_2$  values in some chalcogenide glasses. Subsequently, Petkov and Ewen [27] discussed in more detail application of "Boling's" formula with the Wemple and DiDomenico single oscillator model (W-D) for linear refractive index dispersion and showed that in this case  $n_2$  can be related to some structural parameters of a glass. Independently, Tichý et al. [28] used Miller's generalized rule [20] in combination with linear refractive index and W-D for estimation of  $\chi^{(3)}$  values in some As-Se and Ge-As-Se glasses. For our purposes – a rough estimation of  $\chi^{(3)}$  in amorphous chalcogenides and comparison of  $\chi^{(3)}$  and  $n_2$  values of various oxide glasses and single crystals, we shall combine Miller's generalized rule [20] and low-frequency linear refractive index estimated from Wemple-DiDomenico single effective oscillator model [29]. The reason is that:

(i) Miller's generalized rule is very simple ( $\chi^{(3)} \cong A(\chi^{(1)})^4$ ), this one is physically based, and it gives quite good agreement between predicted and measured  $\chi^{(3)}$  values [19, 30];

(ii) Most of relevant experimental data for amorphous chalcogenides (linear refractive index dispersion measured in visible and/or near infrared spectral region) are in literature interpreted using W-D;

(iii) The parameters of W-D, the dispersion energy ( $E_d$ ) and the energy of effective oscillator ( $E_0$ ) are related to a „chemistry“ of a material that is -  $E_d$  is related to an ionicity, anion valency, coordination number, and  $E_0$  is related to some bond energy or „bond gap“, or band gap [31]. Within W-D, the linear refractive index dispersion is given by relation [29]:

$$n = \left[ 1 + \frac{E_0 E_d}{E_0^2 - (\hbar\omega)^2} \right]^{1/2}, \quad (1)$$

where  $E_d = \beta N_c N_e Z_a$ . For covalent solids  $\beta = 0.34 \pm 0.04$  eV, for ionic solids  $\beta = 0.27 \pm 0.04$  eV,  $N_c$  is the coordination number of the nearest neighbour cation to the anion,  $N_e$  is the total number of valence electrons per anion, and  $Z_a$  is the formal chemical valency of the anion. The linear optical susceptibility for isotropic medium, the case of chalcogenide glasses in the first approximation, is given by relation:

$$\chi^{(1)} = (n^2 - 1)/4\pi, \quad (2)$$

and from (1) and (2) we obtain:

$$\chi^{(1)} = \frac{E_d E_0}{4\pi(E_0^2 - (\hbar\omega)^2)}, \quad (3)$$

which in the limit  $\hbar\omega \rightarrow 0$  ( $n = n_0$ ) gives:

$$\chi^{(1)} = \frac{E_d/E_0}{4\pi}. \quad (4)$$

Using generalized Miller's rule:  $\chi^{(3)} = A(\chi^{(1)})^4$  [19,20] we obtain:

$$\chi^{(3)} = A/(4\pi)^4 (E_d/E_0)^4 = A/(4\pi)^4 (n_0^2 - 1)^4. \quad (5)$$

To estimate whether relation (5) gives reasonable result if applied to experimental  $\chi^{(3)}$  and  $n_0$  values we selected 97 experimental values of  $\chi^{(3)}$  and  $n_0$  of various oxide glasses tabulated in [32]. In Fig.1 the plot  $\log \chi^{(3)}$  versus  $\log (n_0^2 - 1)$  indicates reasonable correlation of experimental  $\chi^{(3)}(n_0)$  values within the relation (5) with the slope 4. Estimated A value is  $A = 1.7 \times 10^{-10}$  (for  $\chi^{(3)}$  in [esu]) which corresponds with A values obtained by Wang [19] for some ionic crystals:  $0.6 \times 10^{-10} < A < 2.5 \times 10^{-10}$ . Adair et al. [33] reported nonlinear refractive index for a large number of optical crystals (oxides and halogenides) measured at near one micrometer using nearly degenerate three-wave mixing method. Using both, the relevant  $E_d$ ,  $E_0$  values taken from [29] and eq. (5), we calculated  $n_2$  values assuming for simplicity [34]:

$$n_2 = 12\pi\chi^{(3)}/n_0. \quad (6)$$

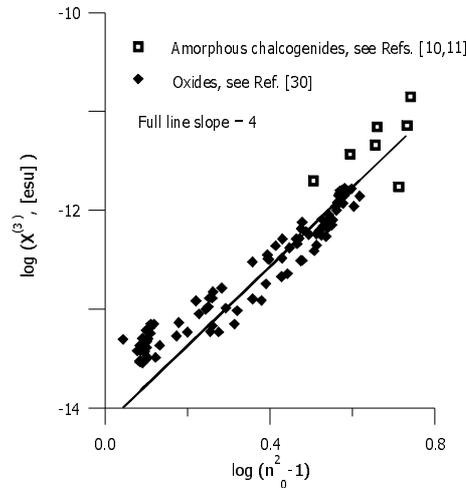


Fig. 1. The dependence of  $\log(\chi^{(3)})$  versus  $\log(n^2-1)$ . Full symbols-experimental values for oxide glasses [30], open symbols - experimental values for chalcogenide glasses [11], solid line- calculated according to Eq.(3).

Following recommendation suggested by Górski et al. [35] we multiplied the  $n_2$  values reported in [33] by a factor of about 2.3. In Fig.2 are shown calculated values of  $n_2$ :  $n_2[\text{esu}] = 2.6 \times 10^{-13} (E_d/E_0)^4/n_0 = 2.6 \times 10^{-13} (n_0^2 - 1)^4/n_0$  versus experimental (2.3  $n_2$ ) values [33]. In the same figure also experimental  $n_2$  values for some chalcogenide glasses recently reported [12] and  $n_2$  values calculated from experimental  $\chi^{(3)}$  values, see [11], using Eq.(6) are shown for reader's convenience. Taking into consideration four order scale of  $n_2$  values the correspondence between the experimental and calculated values is quite good.

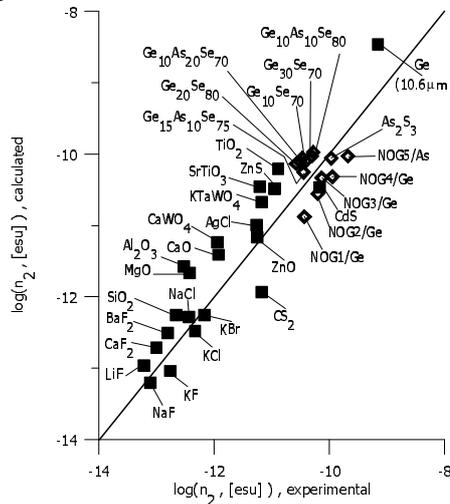


Fig. 2. Calculated  $n_2$  values (Eq.(6)) versus experimental  $n_2$  values, in log-log coordinates. Full symbols, see [31], open symbols - chalcogenide glasses: below line, see [11], above line and  $\text{CS}_2$ , see [12].

In Table 1 are summarized chemical compositions and parameters of W-D single oscillator model for some typical chalcogenide thin films. We selected chemical compositions in rather broad interval in linear refractive index to show possible scale of  $\chi^{(3)}$  values which could be expected for amorphous chalcogenides. From Table 1, it is evident that calculated  $\chi^{(3)}$  values are comparable with experimental  $\chi^{(3)}$  values, see Fig.1 and [10,11]. Hence, using equation (5), the non-linear optical susceptibility can be estimated from linear refractive index ( $n_0$ ) and/or from average oscillator energy ( $E_0$ ) and dispersion energy ( $E_d$ ) of the Wemple-Di Domenico single oscillator model.

Table 1. Chemical composition and the values of the single oscillator energy ( $E_0$ ), the dispersion energy ( $E_d$ ), the linear refractive index ( $n_0$ ), relevant references, calculated non-linear susceptibility ( $\chi^{(3)}$ ) and non-linear refractive index ( $n_2$ ), resp., for various amorphous chalcogenides. [ $n_0 = (E_d/E_0 + 1)^{1/2}$ ;  $\chi^{(3)}$ , calc., see Eq.(5);  $n_2$ , calc., see Eq. (6)].

Chemical composition	$E_0$ [eV]	$E_d$ [eV]	$n_0$	Ref.[No]	$\chi^{(3)}$ , calc. [esu]	$n_2$ , calc. [esu]
As <sub>38</sub> S <sub>62</sub>	5.3	21.8	2.26	36	$1.96 \times 10^{-12}$	$3.26 \times 10^{-11}$
As <sub>40</sub> S <sub>60</sub>	6.0	25.2	2.28	37	$2.13 \times 10^{-12}$	$3.51 \times 10^{-11}$
Ge <sub>4</sub> As <sub>36</sub> S <sub>60</sub>	6.5	25.7	2.23	37	$1.67 \times 10^{-12}$	$2.83 \times 10^{-11}$
Ge <sub>15</sub> As <sub>25</sub> S <sub>60</sub>	6.5	26.4	2.25	37	$1.86 \times 10^{-12}$	$3.11 \times 10^{-11}$
Ge <sub>25</sub> As <sub>15</sub> S <sub>60</sub>	6.4	26.8	2.28	37	$2.1 \times 10^{-12}$	$3.47 \times 10^{-11}$
Ge <sub>36</sub> As <sub>4</sub> S <sub>60</sub>	6.3	27.4	2.31	37	$2.44 \times 10^{-12}$	$3.98 \times 10^{-11}$
As <sub>41</sub> Se <sub>59</sub>	3.59	23.0	2.72	28	$1.15 \times 10^{-11}$	$1.59 \times 10^{-10}$
As <sub>40</sub> Se <sub>60</sub>	4.7	28.0	2.64	37	$8.6 \times 10^{-12}$	$1.23 \times 10^{-10}$
Ge <sub>10</sub> As <sub>30</sub> Se <sub>60</sub>	4.5	23.2	2.48	37	$4.83 \times 10^{-12}$	$7.33 \times 10^{-11}$
Ge <sub>22</sub> As <sub>18</sub> Se <sub>60</sub>	4.44	22.2	2.45	37	$4.27 \times 10^{-12}$	$6.57 \times 10^{-11}$
Ge <sub>36</sub> As <sub>4</sub> Se <sub>60</sub>	4.6	24.2	2.50	35	$5.23 \times 10^{-12}$	$7.88 \times 10^{-11}$
Ge <sub>40</sub> Se <sub>60</sub>	3.7	16.0	2.31	37	$2.39 \times 10^{-12}$	$3.9 \times 10^{-10}$
Ge <sub>10</sub> Sb <sub>30</sub> Se <sub>60</sub>	3.5	20.2	2.60	37	$7.58 \times 10^{-12}$	$1.1 \times 10^{-10}$
Ge <sub>25</sub> Sb <sub>15</sub> Se <sub>60</sub>	4.10	23.3	2.59	37	$7.12 \times 10^{-12}$	$1.04 \times 10^{-10}$
Ge <sub>35</sub> Sb <sub>5</sub> Se <sub>60</sub>	3.8	20.4	2.52	37	$5.67 \times 10^{-12}$	$8.47 \times 10^{-11}$
Ge <sub>50</sub> S <sub>50</sub>	4.12	23.4	2.58	38	$7.11 \times 10^{-12}$	$1.04 \times 10^{-10}$
Ge <sub>33.3</sub> S <sub>66.6</sub>	5.45	19.76	2.15	39	$1.18 \times 10^{-12}$	$2.07 \times 10^{-11}$
As <sub>16</sub> Te <sub>84</sub>	2.07	27.04	3.75	2	$1.99 \times 10^{-10}$	$2.0 \times 10^{-9}$
As <sub>30</sub> Te <sub>70</sub>	2.25	28.22	3.68	2	$1.68 \times 10^{-10}$	$1.72 \times 10^{-9}$
As <sub>40</sub> Te <sub>60</sub>	2.325	28.05	3.61	2	$1.45 \times 10^{-10}$	$1.51 \times 10^{-9}$
Te	2.3	27	3.43	40	$1.30 \times 10^{-10}$	$1.42 \times 10^{-9}$

### 3. The correlation between $n_2$ and the optical gap

It was suggested by Wemple and DiDomenico [29] and by Ke. Tanaka [41] that W-D effective oscillator energy ( $E_0$ ) relates to the optical gap ( $E_g$ ). According to W-D the relation between  $E_g$  and  $E_0$  can be expressed in the form  $E_0 \approx 1.5 E_t$ , where  $E_t$  is the lowest direct band gap. According to Ke. Tanaka for amorphous chalcogenides it is approximately valid  $E_0 \approx 2 E_g$ . For a broad interval in  $E_g$  values:  $0.18$  (InSb)  $\leq E_g$  (eV)  $\leq 13.6$  eV (LiF) we examined the empirical correlation between  $E_g$  and  $E_0$ , see Fig. 3.

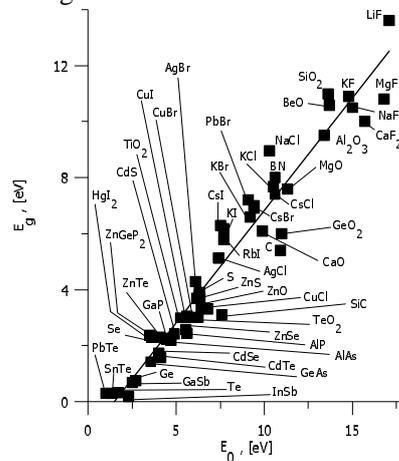


Fig. 3. The empirical correlation between  $E_g$  values and  $E_0$  values for various solids. Full symbols- experimental values, full line - empirical correlation:  $E_g = 0.8E_0 - 1.2$ . The  $E_0$  values were taken from Refs.[29,40], the  $E_g$  values were taken from Refs.[42,43].

From this figure it is evident that within examined scale of  $E_g$ ,  $E_0$  values, the correlation  $E_0(E_g)$  can be expressed in the form:  $E_0 = 1.25 (1.2 + E_g)$ , and hence really  $E_0 \sim E_g$ . Using this proportionality and relations (5,6) one obtains approximately  $n_2 \sim B/E_g^4$ . Fortunately the errors introduced by neglecting the  $E_d$  changes with the chemical composition and taking simple proportionality  $E_0 \sim E_g$  cancels each other and consequently the correspondence between  $n_2$  and  $E_g^{-4}$  shown in  $\log n_2$  versus  $\log E_g$  coordinates in Fig.4 is quite reasonable. The full line in Fig.4 corresponds to the semiempirical relation  $n \sim B/E_g^4$  for  $B = 1.26 \times 10^{-9}$  [esu eV<sup>4</sup>] where  $n_2$  is in esu units and  $E_g$  is in eV units. The region of predicted and also measured  $n_2$ ,  $E_g$  values for chalcogenide glasses is marked by shaded eclipse.

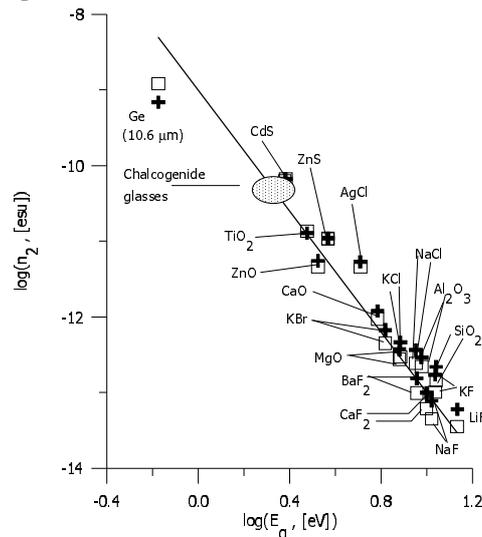


Fig. 4.  $\log(n_2)$  versus  $\log(E_g)$  dependence. The  $n_2$  values multiplied by 2.3, see text, taken from Ref.[33] are marked by +, the  $n_2$  values calculated using Eqs.(5,6) are marked by  $\square$ , the full line with the slope = 4 corresponds to  $n_2 \cong B/E_g^4$ . The  $E_g$  values are taken from Ref. [43]. The shaded eclipse indicates the region of amorphous chalcogenides.

#### 4. Conclusion

Simple semiempirical relation based on generalized Miller's rule allows an estimation of non-linear susceptibility ( $\chi^{(3)}$ ) and non-linear refractive index ( $n_2$ ) from linear refractive index and/or from the dispersion energy and the energy of effective oscillator of the Wemple-Di Domenico model. For a number of various glasses and optical crystals suggested relation indicates reasonable correspondence between calculated and experimental  $\chi^{(3)}$  and  $n_2$  values. Its application to chalcogenide glasses confirms that these materials could reach, in agreement with experimental results [10-14], rather high  $\chi^{(3)}$ ,  $n_2$  values. For a broad range of  $n_2$  and  $E_g$  values the semiempirical relation was found in the form  $n_2(\text{esu}) \cong B/E_g^4$ . Amorphous and glassy chalcogenides have rather high values of  $n_2$  and, hence, they are very interesting materials for some non-linear optical applications.

#### Acknowledgements

Supported partly by projects: MSM 253100001, Key project ASCR 2050602.

#### References

- [1] G. Lucovsky, J. P. Neufville, F. L. Galeener, Phys. Rev.B **9**, 1591 (1974).
- [2] M. Závětová, B. Velický, V. Vorlíček, Solar Energy Materials **8**, 33 (1982).
- [3] Ke. Tanaka, Rev. Solids State Sci., **4**, 641 (1990).
- [4] K. Weiser, R. J. Gambino, J. A. Reinhold, Appl. Phys. Lett. **22**, 48 (1973).
- [5] R. Grigorovici, A. Vancu, L. Ghita, J. Non-Cryst. Solids **59-60**, 909 (1983).

- [6] M. D. Reichtin, A. R. Hilton, D. J. Hayes, *J. Electron. Mater.* **4**, 347 (1975).
- [7] M. Vlček, L. Tichý, J. Klikorka, A. Tříska, *J. Mater. Sci.* **4**, 2508 (1989).
- [8] B. G. Aitken, R. S. Quimby, *J. Non-Cryst. Solids* **213-214**, 281(1997).
- [9] P. Němec, B. Frumarová, M. Frumar, J. Oswald, *J. Phys. Chem. Solids* **61**, 1583 (2000).
- [10] H. Nasu, Y. Ibara, K. Kubodera, *J. Non-Cryst. Solids* **110**, 229 (1989).
- [11] H. Nasu, K. Kubodera, M. Kobyashi, M. Nakamura, K. Kaniya, *J. Amer. Cer. Soc.* **73**, 1794 (1990).
- [12] T. Cardinal, K. A. Richardson, H. Shin, A. Schutte, R. Beatty, K. L. Foulgoe, C. Meneghini, J. F. Viens, A. Villeneuve, *J. Non-Cryst. Solids* **256/257**, 353 (1999).
- [13] F. Smektala, C. Quemard, V. Couderac, A. Barthélémy, *J. Non-Cryst. Solids* **274**, 232 (2000).
- [14] C. Quémard, F. Smektala, V. Ceudere, A. Barthélémy, J. Lucas, *J. Phys. Chem. Solids* **62**, 1435 (2001).
- [15] M. Sheik-Bahae, A. A. Said, E. W. van Stryland, *Opt. Letters* **14**, 955 (1989).
- [16] D. W. Hall, N. F. Borelli, W. H. Dumbaugh, M. A. Newhouse, D. L. Weldman, *Proc. Symp. Nonlinear. Opt.*, Troy, p.293, 1988
- [17] J. Phillips, J. van Vechten, *Phys. Rev.* **183**, 709 (1969).
- [18] B. Levine, *Phys. Rev. Letters* **22**, 757 (1969).
- [19] C. Wang, *Phys. Rev. B* **2**, 2045 (1970).
- [20] J. J. Wyne, *Phys. Rev.* **178**, 1295 (1969).
- [21] W. Harrison, *Phys. Rev. B* **8**, 4487 (1973).
- [22] J. Fournier, E. Snitzer, *IEEE J. Quantum Electron.* **10**, 473 (1974).
- [23] N. Boling, A. Glass, A. Owyong, *IEEE J. Quantum Electron.* **14**, 601(1978).
- [24] G. Agrawal, C. Cojan, C. Flytzamic, *Phys. Rev. B* **17**, 776 (1978).
- [25] M. Sheik-Bahae, D. J. Hagan, E. W. Stryland, *Phys. Rev. Letters* **65**, 96 (1990).
- [26] E. Hajto, P. J. S. Ewen, A. E. Owen, *J. Non-Cryst. Solids* **164-166**, 901 (1993).
- [27] K. Petkov, P. J. S. Ewen, *J. Non-Cryst. Solids* **249**, 150 (1999).
- [28] L. Tichý, H. Tichá, P. Nagels, R. Callaerts, R. Mertens, M. Vlček, *Mater. Letters* **39**, 122 (1999).
- [29] S. H. Wemple, Di Domenico, *Phys. Rev. B* **3**, 1338 (1971).
- [30] H. Nasu, J. D. Mackenzie, *Opt. Eng.* **26**, 102 (1987).
- [31] I. Umezu, K. Miyamoto, N. Sakamoto, K. Maeda, *Jpn. J. Appl. Phys.* **34**, 1753 (1995).
- [32] V. Dimitrov, T. Komatsu, *J. Non-Cryst. Solids* **249**, 160 (1999).
- [33] R. Adair, L. L. Chase, S. A. Payne, *Phys. Rev. B* **39**, 3337 (1989).
- [34] W. L. Smith, in M. J. Weber (Ed.), *Handbook of Laser Science and Technology*, Chemical Rubber Co., Boca Raton, 1986, Vol.3, part 1, p.259.
- [35] P. Górski, M. Kin, W. Kucharczyk, *Phys. Status Solidi (b)* **197**, 265 (1996).
- [36] L. Tichý, H. Tichá, K. Handlíř, *J. Mater. Sci.* **23**, 229 (1988).
- [37] E. R. Skordeva, *J. Optoelectron. Adv. Mat.* **1**, 43 (1999).
- [38] I. Gregora, B. Velický, M. Závětová, *Phys. Status Solidi (b)* **104**, K95 (1981).
- [39] E. Marquez, A. M. Bernal-Oliva, J. M. Gonzales-Leal, R. Prieto-Alcón, J. C. Navarro, D. Minkov, *Phys. Scripta*, **60**, 90 (1999).
- [40] S. H. Wemple, *J. Chem. Phys.* **67**, 2151 (1977).
- [41] Ke. Tanaka, *Thin Solid Films*, **66**, 271 (1980).
- [42] T. S. Moss, *Phys. Status Solidi (b)*, **131**, 415 (1985).
- [43] W. H. Strehlow, E. L. Cook, *J. Phys. Chem. Ref. Data*, **2**(No1), 163 (1973).