

THEMALLY INDUCED STRUCTURAL TRANSFORMATIONS IN CVD TRANSITION METAL OXIDE THIN FILMS

K. Gesheva^{a*}, A. Cziraki^b, T. Ivanova^a, A. Szekeres^c

^aCentral Laboratory of Solar Energy and New Energy Sources of the Bulgarian Academy of Sciences, Blvd. "Tzarigradsko Chaussee" 72, 1784 Sofia, Bulgaria

^bEotvos Lorand University, Faculty of Solid State Physics, Pazmany Peter setany 1, 1117 Budapest, Hungary

^cInstitute of Solid State Physics of the Bulgarian Academy of Sciences, Blvd. "Tzarigradsko Chaussee" 72, 1784 Sofia, Bulgaria

Thin films of molybdenum, tungsten and mixed MoO₃-WO₃ oxides, obtained by atmospheric pressure chemical vapor deposition, were investigated by Raman and XRD measurements. All of the films were prepared using identical technological parameters, and were annealed at 400 °C. Analysis of the Raman and XRD spectra for the three types of film shows that molybdenum oxide films crystallized as orthorhombic MoO₃, with a small fraction of MoO_{2.89} sub-oxide, while the annealed tungsten oxide films were triclinic WO₃ with less than 2 % of the sub-oxide of W₂₀O₅₈. The Raman spectra of the mixed MoO₃-WO₃ films indicated the crystalline phases of molybdenum and tungsten oxides in an amorphous network. In addition, the XRD results showed that the structure of the mixed oxides followed the crystallization of WO₃, and Mo atoms were suggested to substitute for W in the WO₃ lattice.

(Received December 9, 2004; accepted January 26, 2005)

Keywords: CVD thin films, Electrochromic metal oxides, Vibrational properties, XRD analysis

1. Introduction

Transition metal oxides are largely studied with respect to their electrochromic properties, for applications as smart windows and display devices. Molybdenum trioxide, compared to tungsten trioxide, has a comparatively lower coloration efficiency. However, the closer position of its optical absorption peak to the human eye sensitivity peak makes this material a very attractive electrochromic material [1,2]. Mixed films based on W/Mo oxides are expected to exhibit an enlarged optical absorption band. This could originate from increased electron transitions between the two types of metallic site in the two-component material.

Investigations of metal oxide films are connected with their application as the active layers in electrochromic devices, mainly smart windows. A very important feature of the films is their optimal structure (crystalline orthorhombic with a unique layered structure of the MoO₃ film and an amorphous, porous structure in the cases of WO₃ and mixed MoO₃-WO₃ films), all permitting easily intercalation of ions. For an electrochromic device, a high optical transmittance of the active metal oxide layer is very important. To improve the initial transmittance, structural changes of the films are necessary. This is done by post-deposition annealing at temperatures of 300-500 °C [3-5].

In this investigation, the structure of MoO₃, WO₃ and mixed MoO₃-WO₃ oxide thin films after annealing at 400 °C was studied by Raman and XRD methods. The temperature of 400 °C was

* Corresponding author: kagesh@phys.bas.bg

chosen since it is compatible with glass substrate technology and is optimal for a suitably porous film structure.

2. Experimental details

Films of MoO₃, WO₃ and mixed MoO₃-WO₃ oxide were deposited on conductive glass (Donnelly type SnO₂:Sb) and Si substrates, by atmospheric pressure pyrolytical decomposition of the corresponding hexacarbonyls, W(CO)₆, Mo(CO)₆, or their suitable physical mixtures. The sublimator temperature of 90 °C was controlled to an accuracy of ± 1°C. The selected flow rate of argon through the sublimator assured a constant supply of precursor vapor. Oxygen entered separately into the reactor. The chosen ratio of flow rates of argon to oxygen was 1:32. The deposition temperature was 200 °C. This was a crossing point of the deposition temperatures for the three kinds of metal oxide film [5]. The deposition time was kept constant (40 min) and because of the different growth rates the film thickness was about 300 nm for MoO₃, 120 nm for WO₃ and 400 nm for mixed oxide films. Apparently, the presence of two components in the deposition chemical reactions changes the growth kinetics, leading to an increased growth rate of the mixed oxide films at a low CVD temperature (200°C), at which pure WO₃ grows very slowly [3].

After deposition, the films were annealed in air for 1 hour at 400°C. This temperature was selected because of the glass substrates, used also for practical applications in electrochromic windows. Moreover, some cracking effects caused by higher annealing temperatures were observed in the films [2].

For films deposited on silicon substrates, Raman measurements were carried out using a Dilor XY triple monochromator with a multichannel optical detector. During recording of the Raman spectra in the range of 100-1100 cm⁻¹, the intensity of the He-Ne laser ($\lambda = 632.8$ nm) excitation was controlled to an accuracy of 1%. The excitation power on the sample surface was 0.5 mW, and the excitation spot diameter was 3 μ m. Additional information about the crystalline phases was obtained for films deposited on conductive glass substrates, applying X-ray diffractometry using Philips X'Pert equipment. The crystalline phases were identified using data from the ASTM system [6].

3. Results and discussion

The development of crystalline phases in the CVD metal oxide films, due to thermal annealing, was analyzed by large-angle XRD measurements. The results are summarized in Fig. 1, where the XRD patterns for the MoO₃, WO₃ and MoO₃-WO₃ films are given. The appearance of sharp peaks above the broad amorphous one shows that all three kinds of sample contain crystalline phases. However, it is difficult to separate the crystalline part from the amorphous one, because the structure of the glass substrates is also amorphous.

In the case of the MoO₃ films, orthorhombic molybdenum trioxide and triclinic MoO_{2.89} sub-oxide were identified among the crystalline phases (Fig.1). From the analysis of the ratio of the peak intensities, it was established that the orthorhombic MoO₃ crystallites prefer to grow in the <010> direction. The sub-oxide percentage, however, could not be determined because the corresponding X-ray lines coincided with the strongest XRD peak of the crystalline SnO₂ used as a conductive layer on the glass substrate. This SnO₂ peak appeared in all the XRD spectra. The larger part of the crystalline phases in the tungsten oxide films consisted of triclinic WO₃, and there was less than 2 % of W₂₀O₅₈ sub-oxide (Fig. 1). The XRD pattern of the mixed MoO₃-WO₃ film was very similar to that of the WO₃ film. Although the peak positions were the same, the peak intensity was strongly enhanced, because W atoms are partly substituted by Mo atoms. Obviously, in the mixed oxides, (W,Mo)O₃ crystallites are present with the same triclinic modification found for the pure WO₃ film. The Mo atoms replace some of the W atoms in the triclinic WO₃ lattice.

Our previous Raman investigations have shown that films in the as-deposited state are predominantly amorphous [2]. After annealing at 400 °C, the structure of these films underwent different degrees of crystallization, as shown in Fig. 2. The Raman spectra of the WO₃ and MoO₃-

WO₃ films indicate an amorphous-like structure, with superimposed peaks revealing some degree of crystallization. The spectrum of the annealed MoO₃ film corresponds to a thoroughly crystalline structure. In the spectrum of the MoO₃ film, the peaks in the range 600-1000 cm⁻¹ and below 400 cm⁻¹ down to 120 cm⁻¹ are attributed respectively to the stretching and bending vibrations of Mo-O bonds in the orthorhombic modification [7]. A detailed study of the effect of different annealing temperatures on the MoO₃ film structure has been presented elsewhere [1].

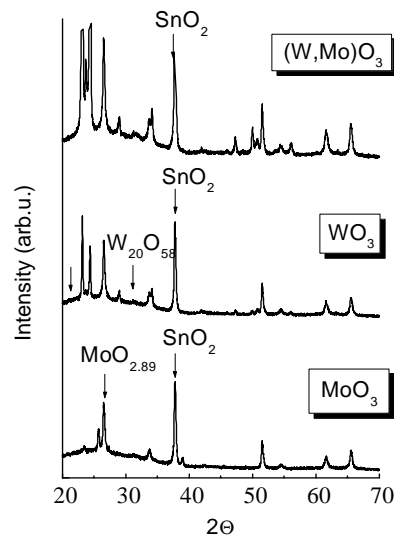


Fig. 1. XRD spectra for MoO₃, WO₃ and mixed MoO₃-WO₃ oxide films deposited on glass substrates at 200 °C and annealed at 400 °C. Most of the peaks are specific for the corresponding trioxides. The non-specific peaks are pointed out in the spectra.

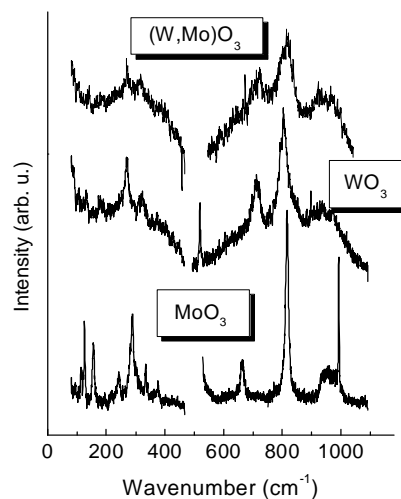


Fig. 2. Raman spectra of MoO₃, WO₃ and mixed MoO₃-WO₃ oxide films, deposited on Si substrates at 200 °C and annealed at 400 °C.

The peaks in the spectrum of the WO₃ film at 717 and 803 cm⁻¹ are connected with the W-O stretching vibrational bands [8]. The similar position of these Raman peaks is related to the monoclinic modification [9]. WO₃ in crystalline form is often cited as either having a monoclinic, orthorhombic or triclinic modification [10]. This is because WO₃ in these modifications has main diffraction peaks at one and the same d-spacings. However, the XRD spectra clearly show a triclinic modification, as discussed above.

In the Raman spectrum of the MoO₃-WO₃ mixed oxide film, there was a very broad band in the range 600-1000 cm⁻¹, centered at around 800 cm⁻¹. This band is also characteristic of the annealed WO₃ film, and it indicates that the mixed film structure was basically a tungsten trioxide amorphous framework. The peaks superimposed over this band are attributed to the crystalline phases of the pure components MoO₃ and WO₃. The peak at around 727 cm⁻¹ is related to W-O stretching vibration bonds [11]. The most pronounced peak at 817 cm⁻¹ could be assigned to the stretching vibration of the Mo-O-Mo groups. The peak at 940 cm⁻¹ is related to the double Mo=O bond vibration in the orthorhombic modification of MoO₃ [7]. These results are in a good agreement with the XRD data, and present additional proof that Mo atoms substitute for W atoms in the WO₃ network.

Additional information about the effect of annealing on these materials is the optical bandgap, E_g, data from ellipsometric measurements, described in detail elsewhere [2,12]. It is empirically established that structural transformation from the amorphous to the crystalline phase causes an increase in the energy band gap for MoO₃ [1,13,14] and a decrease for WO₃ [13,15]. For the annealed MoO₃ films, the E_g value of 2.9 eV is related to crystalline MoO₃. The value of 3.1 eV for the annealed WO₃ films is a characteristic of the amorphous structure of this material. Almost the same value is obtained for mixed MoO₃-WO₃ films (3.0 eV), giving new evidence that the mixed films are predominantly amorphous and are similar to tungsten trioxide.

4. Conclusions

Raman and XRD data analysis showed that the annealing of CVD W/Mo based metal oxide thin films at 400°C leads to crystallization of MoO₃ films in an orthorhombic modification. In the case of WO₃, crystallization occurs in the triclinic modification. For mixed MoO₃-WO₃ oxide films, the structure was dominated by amorphous tungsten trioxide, including the crystalline phases of MoO₃ and WO₃, and the Mo atoms are suggested to substitute for W atoms.

References

- [1] B. W. Faughnan, R. S. Grandall, *App.Phys. Lett.* **31**, 834 (1977).
- [2] K. A. Gesheva, A. Szekeres, T. Ivanova, *Solar Energy Mater. & Solar Cells* **76**, 563 (2003).
- [3] D. Gogova, K. Gesheva, A. Szekeres, M. Sendova-Vassileva, *Phys. Stat. Sol. (a)* **176**, 969 (1999).
- [4] K. A. Gesheva, T. Ivanova, A. Szekeres, A. Maksimov, S. Zaitzev, *J. Phys. IV France* **11**, Pr3-1023 (2001).
- [5] T. Ivanova, K.A. Gesheva, A. Szekeres, A. Maksimov, S. Zaitzev, *J. Phys. IV France* **11**, Pr3-385 (2001).
- [6] Index to the Powder Diffraction File 2000, published by the Joint Committee on Powder Diffractions Standards.
- [7] L. Seguin, M. Figlarz, R. Cavagnat, J. C. Lassegues, *Spectrochimica Acta A* **51**, 1323 (1995) and references therein.
- [8] T. Maruyama, S. Arai, *J. Electrochem. Soc.* **141**, 1021 (1994).
- [9] M.F. Daniel, B. Desbat, J.C. Lassegues, B. Gerand, M. Figlarz, *J. Solid State Chem.* **67**, 235 (1987).
- [10] A. Donnadiou, in *Large-Area Chromogenics: Materials and Devices for Transmittance Control*, Eds. C.M. Lampert and C.G. Granqvist, SPIE Opt. Eng. Press, Bellingham 1990 (Vol.IS4, p.191.)
- [11] T.A. Taylor, H.H. Patterson, *Applied Spectroscopy* **48**, 674 (1994).
- [12] A. Szekeres, T. Ivanova, K. Gesheva, *J. Solid State Electrochem.* **7**, 17 (2002).
- [13] C. G. Granqvist, *Solar Energy Materials & Solar Cells* **60**, 201 (2000).
- [14] C. Julien, A. Khelifa, O. M. Hussain, G. A. Nazri, *J. Crystal Growth* **156**, (1995) 235.
- [15] A. Szekeres, D. Gogova, K. Gesheva, *J. Crystal Growth* **198/199**, 1235 (1999).